Layered Hydroxide Metal Acetates (Metal = Zinc, Cobalt, and Nickel): Elaboration via Hydrolysis in Polyol Medium and Comparative Study

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Layered hydroxide metal acetates (metal = zinc, cobalt, and nickel) have been prepared by a new route belonging to the *chimie douce* method. This novel method involves the hydrolysis, in polyol medium, of in situ-formed complexes supposed to be alkoxyacetates. These layered hydroxide metal acetates present poorly ordered character, and their X-ray patterns have features typical of lamellar compounds with turbostratic disorder. Their chemical formula was established to be $M(OH)_{2-x}(CH_3COO)_x \cdot nH_2O$ with (x, n) = (0.42, 0.31), (0.38, 0.53), and (0.40, 0.63) for Zn, Co, and Ni, respectively. The layered hydroxide nickel acetate has a classical brucite structure with a random substitution of some hydroxyl groups by acetate groups. The layered hydroxide zinc and cobalt acetates are isomorphous with the hydrozincite structure, in which cations are located in both octahedral and tetrahedral sites. The acetate anion behaves as a unidentate ligand in LHS–Ni and LHS–Co, where LHS indicates layered hydroxide salt, and is intercalated as a free anion in the zinc compound. The dehydration is a reversible topotactic process for LHS–Ni and Co, whereas it is a destructive process in the case of LHS–Zn.

Introduction

Several works in the past decade¹⁻⁵ have shown that a great variety of inorganic compounds (metal, oxides) can be easily obtained in polyols because of numerous properties of such solvents, including high boiling point (up to 250 °C) and complexing, reducing, and surfactant properties, in addition to their amphiprotic character.

General principles that emerge from these studies show that the main chemical reactions occurring in polyols are reduction and hydrolysis. Competition between these reactions is easily controlled by adjustment of the hydrolysis ratio. The absence of water favors reduction, leading to micron or submicron metal particles with tailored size and morphology.⁶ On the contrary, the presence of water inhibits reduction and enhances hydrolysis, so that the corresponding oxides are obtained via inorganic polymerizations.⁷ Molecular precursors subject to these reactions are supposed to be in situ-formed alkoxides or intermediate alkoxy—salt complexes. Such species have clearly been evidenced and structurally characterized for zinc acetate dissolved in diethyleneglycol or ethyleneglycol.^{8,9} Therefore, this novel method for elaborating metal oxides can be considered as an intermediate route between the two well-known sol-gel processes that involve hydrolysis reactions of an inorganic salt in aqueous solution or of a metal alkoxide in organic solvent (alcohol).¹⁰

The aim of this work is two-fold. First, it shows that this novel route can be extended to elaborate layered hydroxide metal acetates (metal = Ni, Co, and Zn), which have, up to now, been prepared in aqueous solution.^{11–14} The second goal of this work is to bring new insights into the thermal behavior and morphological and structural features of these materials. Indeed, as will be shown in this paper, the cited previous works have led to incomplete and arguable characterizations. This is because of the poorly crystallized character of such compounds and the limited experimental proofs. We demonstrate here that a more accurate characterization can be obtained by the use of appropriate and complementary techniques: ionic chromatography, temperature-resolved X-ray diffraction and IR spectroscopies,

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Table 1. Preparation of Layered Hydroxide Acetate Metal (M = Zn, Co, and Ni)

reactant	conc (M)	solvent	dissolving temp (°C)	h	precipitation temp (°C)
ZnAc ₂ ·2H ₂ O	0.5 0.5 0.1	DEG 1,2-propanediol ethanol	130 112 76	>2 >2 >2 >2	RT RT RT
CoAc ₂ ·4H ₂ O	0.1 0.1 0.05	DEG 1,2-propanediol ethanol	160 150 56	${}^{\geq 26}_{\geq 26}_{\geq 26}$	60 60 60
NiAc ₂ •4H ₂ O	0.1 0.1 0.05	DEG 1,2-propanediol ethanol	110 140 61	$\geq 4 \\ \geq 4 \\ \geq 4$	60 60 60

TG coupled to mass spectrometry analysis, and UV spectroscopy. Furthermore, whereas in previous works these compounds have been separately studied, our comparative investigation evidences significant differences between these three solids despite their similar general characteristics.

It is worth noting that these compounds are of interest in the field of materials. For instance, the hydroxide cobalt and nickel acetates belong to a family of layered compounds showing very interesting magnetic properties.^{15–17} In addition, applications in the fields of anionic exchange and electrode materials can be expected.

Experimental Section

Synthesis. The hydrated metal acetates $Zn(CH_3COO)_2$ · $2H_2O$, $Co(CH_3COO)_2$ · $4H_2O$, $Ni(CH_3COO)_2$ · $4H_2O$, polyols [diethyleneglycol (DEG), 1,2-propanediol], and ethanol were purchased from Prolabo and used without any further purification.

The general procedure involves dissolution of an acetate salt in a given volume of polyol (250 mL), followed by addition of water at room temperature (RT) and hydrolysis. The precipitated product is washed several times with ethanol and dried in an oven.

The main factors governing the formation of the desired compounds are the hydrolysis ratio, defined as $h = nH_2O/nM$ (M = Ni, Co, Zn), and the metal concentration. The optimal experimental conditions required are listed in Table 1.

As shown in this table, the compounds can be obtained for a large range of hydrolysis ratios. Under the lower limit, the solution remains limpid, and no precipitation is observed. It should be noted that a ratio *h* equal to 250 (a mixture of 250 mL of polyol and 110 mL of water) for a metal concentration C = 0.1 M has been retained to synthesize the studied compounds. This approach will be justified below.

Metal concentrations used in ethanol are low, as salts barely dissolve in this solvent. In the other solvents, nickel and cobalt concentrations higher than 0.1 M led to a mixture of the desired compound and a side phase. For the three metals, the lowest yield was observed for ethanol, and the highest for DEG. Thus, this latter solvent has been retained to elaborate the studied compounds.

Chemical Analysis. Elemental analysis of metal, carbon, and hydrogen was performed at the Vernaison Analysis Centre of CNRS. Zinc, cobalt, and nickel ions were also titrated by EDTA using Eriochrome black T for the two former and alcoholic dithizone for the latter as the indicators. The total base content was estimated by dissolving a known quantity of the sample in an excess of hydrochloric acid and back-titrating the excess of acid against standard sodium hydroxide



Figure 1. X-ray diffraction patterns of (a) LHS–Zn, (b) LHS–Co, and (c) LHS–Ni (Cu K α).

using a pH meter. The acetate content was determined by ionic chromatography with a DIONEX DX-100 ion chromatograph equipped with AG-4 and AS4–SC columns.

Characterization. X-ray diffraction was carried out with two types of diffractometer (a Siemens D5000 diffractometer for RT measurements and a vertical Phillips PW1050 diffractometer, in a Pt–Rh40% sample holder, for temperature-resolved diffraction), in the 2θ range $5-100^{\circ}$, with a step of $0.03^{\circ}(2\theta)$ and a step duration of at least of 17 s.

IR spectra were recorded with a FTIR Perkin-Elmer 1750 spectrophotometer in transmittance mode between 4000 and 500 cm⁻¹ with a resolution of 4 cm⁻¹ and with at least 20 acquisitions. The samples were prepared in KBr pellets. Temperature-resolved IR spectroscopy was performed in diffuse reflectance mode under nitrogen in a Harrick praying mantis apparatus.

Differential thermal and thermogravimetric analyses (DTA and TGA) were carried out with a heating rate of 1 $^{\circ}$ C min⁻¹ in an alumina crucible under oxygen in a Setaram TG92–12 thermal analyzer. Thermal analysis (Setaram balance TG DSC111) coupled with mass spectrometric analysis (Leybold Inficom H 300 CIS spectrometer) were carried out under argon up to 700 $^{\circ}$ C in a platinum crucible with a heating rate of 5 $^{\circ}$ C min⁻¹.

UV-visible-NIR diffuse reflectance spectra were recorded between 200 and 2500 nm on a Cary 5E spectrophotometer equipped with an integration sphere coated with pressed polytetrafluoroethylene (PTFE).

Electron microscopy and diffraction studies were conducted on a JEOL-100 CX II microscope.

Results

X-Ray and Morphological Characterizations. Regardless of the solvent used (DEG, 1,2-propanediol, or ethanol), the X-ray diffraction patterns show the formation of the same compound for a given metal with almost identical interlayer distances. All of the X-ray patterns have features typical of incompletely ordered lamellar compounds related to the brucite structure (Figure 1): sharp reflections in the range of small Bragg angles and broad and asymmetric reflections in the range of high angles. Such features can be explained on the basis of stacking order of brucite-like sheets parallel and equidistant but twisted and/or translated against each other. This turbostratic behavior has been observed for α -Ni(OH)₂, for α -Co(OH)₂,^{18–20} and for the

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Figure 2. TEM micrographs of (a) LHS–Ni, (b) LHS–Zn, and (c) LHS–Co.

relatively wide family of layered double hydroxides (LDH) $[M^{2+}_{1-x}M'^{3+}_{x}(OH)_2]^{x+}[A^{m-}_{x/m} \cdot nH_2O]^{.21}$ Similarly, the studied compounds will be called layered hydroxide

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Figure 3. Electron diffraction patterns of (a) LHS–Ni and (b) LHS–Co.

salts (LHSs) and refered to as LHS–M (M = Ni, Co, and Zn).

Electron microscopy reveals significantly different morphologies for the three LHSs (Figure 2). On one side, LHS-Ni and LHS-Zn exhibit characteristic features of turbostratic materials: they appear as aggregates of thin crumpled sheets without any definite shape. LHS-Zn decomposes under electron beam, and LHS-Ni gives a pattern of rings (Figure 3 a) corresponding to hk0 reflections. LHS-Co is significantly more crystallized. It consists of an aggregate of thin hexagonal platelike particles with a diameter close to 0.5 μ m. Electron diffraction shows that hexagonal platelets are mono crystals lying on the (001) plane, as all observed reflections can be explained as *hk*0 reflections (Figure 3b). This unexpected morphology has been observed for a blue cathodically deposited cobaltous hydroxide, α -Co(OH)2.22

The X-ray patterns have been indexed in the hexagonal system (Table 2), in agreement with electron microscopy results discussed above and general predictions of X-ray patterns of turbostratic materials.²³ Sharp

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Table 2. Cell Parameters and X-ray Diffraction Patterns of LHS-M (M = Zn, Ni, and Co)

	$\frac{M = Zn}{3.138(1)} \\ 14.75(1)$			$\frac{M = Co}{\begin{array}{c} 3.118(2) \\ 12.84(2) \end{array}}$			M = Ni 3.106(3) 10.64(3)		
a (Å) c (Å)									
hkl	d _{obs} (Å)	$d_{ m calc}$ (Å)	$I_{\rm rel}$	d _{obs} (Å)	$d_{ m calc}$ (Å)	$I_{\rm rel}$	d _{obs} (Å)	$d_{ m calc}$ (Å)	$I_{\rm rel}$
001	14.68	14.75	100	12.90	12.84	100	10.54	10.64	100
002	7.36	7.38	5	6.43	6.42	12	5.30	5.32	19
003	4.91	4.92	4	4.29	4.28	7	3.55	3.55	6
004	_	_	_	3.211	3.211	1	_	_	_
100	2.717	2.718	24	2.700	2.701	12	2.680	2.690	19
110	1.569	1.569	13	1.560	1.559	8	1.553	1.553	13
200	1.359	1.359	3	_	_	-	1.346	1.345	1

Table 3. Calculated and Observed Chemical Results for LHS-M (M = Zn, Ni, and Co)

		Ac/M	%M		%C		%	Н	% mass loss	
compound	formula	exp	exp	calc	exp	calc	exp	calc	exp (ATG)	calc
LHS-Zn	Zn(OH) _{1.58} (Ac) _{0.42} •0.31H ₂ O	0.42	51.78	53.32	9.27	8.22	2.69	2.82	32.15	33.62
LHS-Co	Co(OH) _{1.62} (Ac) _{0.38} •0.53H ₂ O	0.38	48.00	49.76	9.05	7.70	3.26	3.23	31.90	32.23
LHS-Ni	Ni(OH) _{1.60} (Ac) _{0.40} •0.63H ₂ O	0.40	46.53	48.58	9.55	7.94	3.26	3.36	39.14	38.18

reflections were explained as 00*l*, and asymmetric reflections as *hk*0. The parameters in the basal plane that represent the metal-metal distances are very close to those of the corresponding β -M(OH)₂ compounds (for M = Zn, Co, and Ni, d_{M-M} = 3.194, 3.173, and 3.114 Å, respectively).^{24–26}

The 001 reflection broadening for LHS-Ni is higher than that for the two other compounds. The crystallite mean size along the c axis (50 Å) inferred from the Scherrer formula corresponds to a very small number of layers per crystallite: 5 compared to about 15 layers for LHS-Zn and 50 layers for LHS-Co.

Chemical Analysis and Formula. Results obtained from chemical analysis, acido-basic titration, and ionic chromatography (analysis of acetate ion) are summarized in Table 3.

The observed carbon amount is higher than that due to acetate deduced from ionic chromatography. This shows that polyol is still present in the as-synthesized compounds. Two types of interaction can occur: a physical adsorption on the external surface of particles or an intercalation of polyol between the layers, as was observed in earlier works.^{27,28}

Despite the bidimensional character of the hydroxyacetates that are the object of this work, X-ray patterns of the compounds obtained in different solvents clearly show that variations in the nature of the polyol have no effect on the interlamellar distance, which remains constant for a given cation. Furthermore, work in ethanol led to X-ray patterns identical to those obtained in polyol. Accordingly, it appears that the solvent is only adsorbed on the external surface of the crystallites. It should be noted that the amount of adsorbed polyol depends on the hydrolysis ratio. It is as low as 0.03, 0.04, and 0.05 mol of polyol per 1 mol of Zn, Co, and Ni, respectively, when h = 250.

The water amount in the formulation was deduced by subtracting the mass of metal, hydroxide, acetate,



Figure 4. IRFT spectra at room temperature of (a) LHS-Zn, (b) LHS-Co, and (c) LHS-Ni (* = adsorbed polyol)

and polyol from the total mass. Chemical formulas inferred from these results are also reported in Table 3.

The observed total loss was deduced from TGA. The agreement between expected and observed results can be deemed satisfactory if one considers the amount of adsorbed solvent.

Infrared Study. Infrared studies confirm the composition established by chemical and TG-mass spectrometric investigations. They clearly show the presence of acetate, hydroxyl, and water species.

For each compound (Figure 4), at least two bands are observed in the range 1600-1300 cm⁻¹: at 1583 and 1411 cm⁻¹ for Zn and at 1572 and 1417 cm⁻¹ for Co. For Ni, in addition to the two intense bands at 1614 and 1381 cm^{-1} , a third band is observed at 1431 cm^{-1} . If we omit the latter band, which may be due to CH_3 deformation,²⁹ the two bands observed for each compound are the fingerprint of the acetate group. Indeed, the bands v_{as} and v_s of C=O in acetate group lie in the ranges 1620-1570 and 1430-1300 cm⁻¹, respectively.30

The temperature-resolved IR study reveals that the difference $\Delta v = v_{as} - v_s$ increases from 227 and 163 cm⁻¹ to 255 (at 250 °C) and 220 cm⁻¹ (at 200 °C) for LHS-

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Table 4. Evolution of Δv with Temperature for LHS-M (M = Zn, Ni, and Co)



Figure 5. Temperature-resolved IRFT spectra of (a) LHS–Zn, (b) LHS–Co, and (c) LHS–Ni, in the range of 4000–2500 cm⁻¹.

Ni and LHS–Co, respectively, in the temperature range corresponding to the thermal stability of these compounds. For the LHS–Zn compound, this difference (172 cm⁻¹) remains almost constant upon heating (Table 4).

Hydroxyl groups and water are evidenced by the broad bands observed at high frequency, at 3350 and 3400 cm⁻¹ for Zn and Co, respectively, and by two broad bands at 3323 and 3553 cm⁻¹ for Ni. The broadening of these bands is due to hydrogen bond formation.^{31,32} In fact, for Co, the broad adsorption band became resolved into at least two components when the temperature was increased. Lower frequency bands at 3400 and 3323 cm⁻¹ for Co and Ni, respectively, correspond to interlamellar water, as these bands disappeared upon dehydration, as shown in temperature-resolved IR spectra (Figure 5b,c) and as already attributed in the case of α -Ni(OH)₂.³³ This dehydration step is clearly identified from the TG and mass spectrometric analyses. The band at higher frequency is due to the hydroxyl group of the brucite-like layer. For Zn, the band at 3350 cm^{-1} remains unresolved upon heating (Figure 5a). The expected band (1600 cm⁻¹) arising from the bending mode δH_2O is obscured by the strong band at 1500-1600 cm⁻¹ due to the C=O asymmetric stretching mode, as observed in previous works on hydrated metal acetates³⁴ and Cu₂(OH)₃(CH₃COO)·H₂O.³⁵

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Thermal Evolution. Thermal Analysis. Figure 6 represents thermogravimetric and differential thermal analyses (TGA and DTA) conducted under oxygen. Chemical species evolved during thermal treatment have been identified by TGA coupled to mass spectrometry. X-ray diffraction patterns show that the final products of decomposition are NiO, Co₃O₄, and ZnO. Ni and Co compounds have similar thermal behaviors. A first loss due to the departure of water occurs around 60 °C and corresponds to a broad endothermic peak. A second weight loss is observed at higher temperature in the ranges 240-285 °C and 180-240 °C for the Ni and Co compounds, respectively, and corresponds to exothermic peaks. It involves formation of water and CO₂ through the dehydroxylation reaction and combustion of acetate anions. This combustion with a strong exothermic effect probably corresponds to a catalytic oxidation of acetate caused by the presence of Ni and Co oxides.

For Zn, two losses with endothermic peaks are observed. The first loss, at around 90 °C, is due to the departure of intercalate water. The second, at around 120 °C, corresponds to the release of water concomitantly with the dehydroxylation of the layers. A third loss, in the range 250-300 °C, corresponds to the departure of CO₂ through a slow decomposition of the acetate group.

This difference will be discussed later in relation to structural characteristics.

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Figure 6. TGA and DTA curves of (a) LHS–Ni, (b) LHS–Co, and (c) LHS–Zn under oxygen.

Temperature Resolved X-ray Diffraction. When heated, LHS–Co and LHS–Ni on one side and LHS–Zn on the other side exhibit different behavior (Figure 7a–c, respectively). For LHS–Co and LHS–Ni, the 00/ reflections decrease in intensity and become more diffuse, suggesting disorder in the stacking of layers, and the corresponding interlamellar distances collapse from 12.9 and 10.5 Å to 10.2 and 8.2 Å, respectively. This collapse is due to the departure of water, as was shown by the analysis of TGA coupled to mass spectrometry. Conversely, the *hk*0 distances remain almost constant. The decomposition of the Ni and Co compounds occurred, respectively, at 290 and 230 °C, leading to finely divided NiO and Co₃O₄ according to their very broad diffraction peaks.



Figure 7. Temperature-resolved X-ray diffraction patterns of (a) LHS–Co (Fe K_{α}), (b) LHS–Ni and evolution of the distance 001 (Cu K_{α}), and (c) LHS–Zn (* = zinc oxide, ∇ = zinc oxoacetate) (Cu K α).

For LHS–Zn, the 00*l* distances remain constant upon heating, but their intensities become weaker and more diffuse until destruction of the compound at a relatively low temperature (120 °C). The analysis of TGA coupled to mass spectrometry showed that this decomposition is due to the departure of water, including that due to the dehydroxylation process. Indeed, this decomposition leads to the formation of a mixture of zinc oxide and a second phase identified as a zinc oxoacetate by Golgotiu et al.³⁶ This phase undergoes a transformation at higher



Figure 8. UV-visible spectra of LHS-Ni.



Figure 9. UV-visible spectra of LHS-Co.

Table 5. Energies and Assignments of Bands Observed in the UV-Vis Spectra of LHS-M (M = Ni and Co)

compound	transition energy (cm ⁻¹)	assignment	symmetry
LHS-Ni	8064 13141 14903 22747 25840	$\begin{array}{l} v_1\colon {}^{3}T_{2g}(F) \leftarrow {}^{3}A_{2g}(F) \\ v_2\colon {}^{3}T_{1g}(F) \leftarrow {}^{3}A_{2g}(F) \\ {}^{1}E_g(G) \leftarrow {}^{3}A_{2g}(F) \\ {}^{1}T_{1g}(G) \leftarrow {}^{3}A_{2g}(F) \\ v_3\colon {}^{3}T_{1g}(P) \leftarrow {}^{3}A_{2g}(F) \end{array}$	O_h
LHS-Co	6382 8267 15596 and 16883 20267	$\begin{array}{c} v_2 \overset{T_{a_i}}{\to} {}^{4}T_1(\mathbf{F}) \leftarrow {}^{4}A_2(\mathbf{F}) \\ v_1 \overset{O_{h_i}}{\to} {}^{4}T_{2g}(\mathbf{F}) \leftarrow {}^{4}T_{1g}(\mathbf{F}) \\ v_3 \overset{T_{a_i}}{\to} {}^{4}T_1(\mathbf{P}) \leftarrow {}^{4}A_2(\mathbf{F}) \\ v_3 \overset{O_{h_i}}{\to} {}^{4}T_{1g}(\mathbf{P}) \leftarrow {}^{4}T_{1g}(\mathbf{F}) \end{array}$	$T_d \\ O_h \\ T_d \\ O_h$

temperature that leads to zinc oxide.

Hydration-dehydration appears to be a reversible process for LHS-Co and LHS-Ni. Upon exposure to moist air, dehydrated LHS-Co transformed into hydrated form more rapidly (90 min) than LHS-Ni (48 H).

UV-Visible-NIR Reflectance Spectroscopy. The UV-visible spectra of the compounds are shown in Figures 8 and 9. Energies and assignments of the observed bands are given in Table 5.

The nickel compound spectrum shows weak d-d absorption corresponding to Ni(II) in octahedral coordination of a weak ligand. The crystal field strength Δ_{O_h} deduced (8172 cm⁻¹) (the average value obtained from Δ_{O_h} for v_1 and v_2) is close to the value for β -Ni(OH)₂ (Δ_{O_h} = 8600 cm⁻¹) and the Racah parameter B = 1041 cm⁻¹ is in the same range as that observed for β -Ni(OH)₂, B $= 820 - 925 \text{ cm}^{-1}.37$

The cobalt compound spectrum shows d-d absorption corresponding to Co(II) in octahedral coordination and two additional sharp peaks at lower energies due to Co(II) in a tetrahedral coordination geometry. The crystal field strength values ($\Delta_{O_h} = 10$ 334 cm⁻¹ and $\Delta_{T_d} = 3545 \text{ cm}^{-1}$), calculated, respectively, with $v_1(O_h)$ and $v_2(T_d)$ and the Racah parameters (B_{O_h} and $B_{T_{d^h}}$ respectively, 938 and 756 cm⁻¹) are in good agreement with those given in the literature ($\Delta = 9200 \text{ cm}^{-1}$ and $B = 825 \text{ cm}^{-1}$ for $[C_0(OH_2)_6]^{2+38}$ and $\Delta = 3900 \text{ cm}^{-1}$ and $B = 775 \text{ cm}^{-1}$ for ZnO:Co³⁹).

Discussion

A New Preparative Route: Hydrolysis in Polyol Medium. To date, LHS compounds have been prepared via two main routes conducted in aqueous media: partial basic titration at RT and hydrolysis, at elevated temperature under hydrothermal conditions, of a metal salt solution or exchange reactions. Our work describes a new method, not yet reported in the literature, for preparing such materials. This novel route belongs to the general chimie douce method and is closely related to the sol-gel process. It involves several steps, including the dissolution of metal acetate in polyol medium and the in situ formation of complexes (supposed to be alkoxyacetates), which are then hydrolyzed. This novel route is very attractive and offers several advantages. When the work is performed in polyol medium, the precipitation of the hydroxyacetates is easily controlled through adjustments in the hydrolysis ratio for a ratio of acetate/metal equal to 2. In the aqueous method, this step requires a very careful control of the pH by partial titration with NaOH. Furthermore, for LHS-Ni, the synthesis in aqueous solution led to a contamination with carbonate anion. Avoidance of this contamination requires synthesis under hydrothermal conditions at 160 °C. In polyol medium, this is achieved more easily at 60 °C and under atmospheric condition. As has been suggested by Hansen et al., polyols can react with carbonate, leading to the release of CO₂.⁴⁰

Structural Characteristics. Table 6 allows the studied compounds to be compared to similar compounds prepared in aqueous solution.

The two routes led to compounds with similar formulas if we omit the very low amount of carbonate in the LHS-Ni prepared in aqueous medium and the high amount of water for the LHS-Ni obtained under hydrothermal conditions. However, their physical characteristics appear to be slightly different. Interlamellar distances are higher for LHS-Zn and LHS-Ni prepared in polyol than for the corresponding compounds elaborated in aqueous solution. The Δv values [$\Delta v = v_{as}(CO)$] $v_{\rm s}({\rm CO})$] are also different.

X-ray patterns of the three solids are similar to those observed for layered compounds with brucite⁴¹ or hydrozincite⁴² structures. Brucite can be described as a packing of single layers consisting of Mg(OH)₆ edgesharing octahedra, the cohesion between layers being

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Table 6. Characteristics of LHS-M (M = Zn, Co, Ni) Prepared in Polyol and Aqueous Media

compound	formula	interlamellar distance (Å)	$\Delta v (\mathrm{cm}^{-1})$ ($v_{\mathrm{as}} - v_{\mathrm{s}}$)	ref
LHS-Zn	Zn(OH) _{1.58} (Ac) _{0.42} ·0.31H ₂ O Zn(OH) _{1.60} (Ac) _{0.40} ·0.60H ₂ O	14.68 13.8	172 195	this work 13
LHS-Co	$\begin{array}{l} Co(OH)_{1.62}(Ac)_{0.38} {\scriptstyle \cdot } 0.53H_2O \\ Co(OH)_{1.50}(Ac)_{0.50} {\scriptstyle \cdot } 0.50H_2O \end{array}$	12.90 12.8	163 —	this work 12
LHS-Ni	$\begin{array}{l} Ni(OH)_{1.60}(Ac)_{0.40} \cdot 0.63H_2O \\ Ni(OH)_{1.55}(Ac)_{0.43}(CO_3)_{0.01} \cdot 0.20H_2O \\ Ni(OH)_{1.5}(Ac)_{0.5} \cdot 1.4H_2O \end{array}$	10.54 9.3 9.44	230 180 208	this work 11 14

assured by van der Waals forces. β -M(OH)₂ compounds (M = Ni, Co, and Zn) belong to this structural type. The hydrozincite structure has the form of an inorganic triple-layer slab. Indeed, in this structure, a brucite sheet, in which one-quarter of the cation sites are vacant, is sandwiched between two layers formed by two tetrahedral zinc layer sheets placed above and below each octahedral vacant site. This triple-layer slab type is encountered in zinc hydroxychloride Zn₅(OH)₈Cl₂. 1H₂O⁴³ and hydroxnitrate Zn₅(OH)₈(NO₃)₂·2H₂O.⁴⁴ However, these two compounds differ by the nature of the organization of the chemical species in the interlamellar space. For $Zn_5(OH)_8(NO_3)_2 \cdot 2H_2O$, the tetrahedral fourth apex pointing into the interlayer space is occupied by the oxygen of a water molecule with nitrate being intercalated in the gallery as a free ion. For Zn₅(OH)₈-Cl₂·H₂O, chloride ion occupies the fourth apex, and water forms a layer in the interlamellar space. To show both coordinations, these formulas can be written as $[(Zn^{octa})_{1-x}(Zn^{tetra})_{2x}(OH)_2]^{+2x}\cdot 2x(A^-)\cdot nH_2O$ with x = 0.25, the anion-to-metal ratio being equal to 0.4.

The comparative study conducted here reveals that the hydroxyacetates of Ni, Co, and Zn, despite their similar X-ray patterns and general chemical formula, present significant structural differences, which concern mainly the cation coordination concomitantly with the thickness of the inorganic layer (single or triple layers) and the organization of the chemical species (water molecule, acetate anion) in the gallery. These two points will be discussed hereafter.

Cation Distribution and Nature of the Layers. UV characterization shows that the nickel ion is exclusively in an octahedral coordination. This indicates that the inorganic sheets in LHS-Ni are constituted by single layers as in β -Ni(OH)₂.

In the case of LHS-Co and LHS-Zn, the stacking of single layers is insufficient to account for the high interlamellar distances observed in comparison with those of LHS-Ni.

For LHS-Co, UV analysis shows qualitatively the occurrence of Co²⁺ in both octahedral and tetrahedral sites. The absorption bands of the Co^{T_d} , about a hundred times more intense than the octahedral absorption band, excludes any quantitative determination of the ratio Co^{*T*}_d/Co^{*O*_h} by this method.⁴⁵ However, this qualitative result, along with the relatively high interlamellar distance (12.9 Å) and the observed acetate/metal ratio (0.4), is consistent with a structural model with triplelayer slabs, such as observed for Zn₅(OH)₈Cl₂·1H₂O and $Zn_5(OH)_8(NO_3)_2 \cdot 2H_2O$. This conclusion is in qualitative agreement with that reached by Laget et al.¹² for LHS-Co prepared in aqueous solution, with, however, the slightly different formula $[(Co^{octa})_{1-x}(Co^{tetra})_{2x}(OH)_2]^{+2x}$ $2x(Ac^{-}) \cdot nH_2O$ with x = 0.16.

In LHS-Zn, zinc cations have been found in both tetrahedral and octahedral coordinations. Indeed, starting from LHS-Zn, we have exchanged the acetate anion by chloride and nitrate and recovered the Zn₅(OH)₈Cl₂· 1H₂O and Zn₅(OH)₈(NO₃)₂·2H₂O homologues whose structures were described earlier. It should be noted that the inverse exchange is also possible. Indeed, Kawai et al. elaborated the LHS-Zn compound by solid/ liquid exchange between Zn₅(OH)₈Cl₂·1H₂O and zinc acetate solution¹³ and accordingly concluded that LHS-Zn is isostructural with the hydroxychloride. As for LHS-Co, the Zn compound appears to be formed by inorganic triple-layer slabs, in agreement with the high interlamellar distance observed (14.7 Å).

Interlamellar Organization and Structural Model. Regardless of the temperature (up to 280 °C), LHS–Ni elaborated in polyol presents a Δv value higher than 200 cm⁻¹. This indicates a unidentate character for the acetate anion according to Nakamoto rules.³⁰ Such character is consistent with a structural model in which nickel ions are in an octahedral environment forming brucite-like layers. Twenty percent of the anion sites are occupied by acetate species, with the remaining sites being occupied by hydroxyl groups. A very recent work¹⁴ suggested a similar character for the acetate anion for LHS-Ni prepared under hydrothermal conditions. However, our model is in contradiction with two other models proposed in the literature. The first model¹¹ derives from the brucite structure by the occurrence of vacancies in hydroxyl sites, thus giving a cationic sheet $[M(OH)_{2-x}\Box_x]^{+x}$. Electrical neutrality is fulfilled by intercalation of acetate and carbonate anions along with water between the layers. This model is based on Nakamoto rules, which predict a free character for acetate ion that shows a Δv value less than 200 cm⁻¹. Such a conclusion is subject to discussion because several works have shown that Δv may be lowered when the acetate ions are engaged in hydrogen bonds. This was clearly demonstrated for Cu₂(OH)₃(CH₃COO)·H₂O³⁵ for which structural determination has shown that the acetate has a unidentate character despite a Δv value of 140 cm^{-1} . Another model has been invoked, with no experimental proof, to explain the positive charge of the sheets in α-hydroxides.⁴⁶ It consists of partial protonation of the hydroxyl groups according to the equilibrium $M(OH)_2 + xH^+ \Leftrightarrow [M(OH)_{2-x}(H_2O)_x]^{+x}$, with anions still intercalated as free ions between the layers.

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Layered Hydroxide Metal Acetates

Our study evidences for the first time the role of water in the organization of the interlamellar space. When the temperature is increased from RT to 280 °C, a decrease in the lamellar distance by 2.6 Å (from 10.5 to 8.20 Å) is observed. This variation corresponds to the water van der Waals diameter⁴⁷ and thus indicates that this chemical species is intercalated as a molecular layer (Figure 10a). Dehydration is confirmed by the disappearance of the broad band at 3323 cm⁻¹ in the IR spectra, by the mass spectral analysis, and by the increase in Δv of the acetate anions because of the breaking of hydrogen bonds. It should be noted that this study shows that LHS–Ni undergoes a topotactic dehydration process that conserves the lamellar character.

Although LHS–Co and LHS–Zn belong to the hydrozincite structural type, temperature-resolved IR and X-ray studies revealed significant differences in the organization of chemical species between the layers.

In LHS-Co, this organization is similar to that observed for LHS–Ni. Indeed, the increase in Δv from 163 to 220 cm⁻¹ (at 200 °C) upon dehydration shows clearly the unidentate character of acetate. Furthermore, when the temperature is increased from RT to 280 °C, a decrease in the lamellar distance by 2.6 Å is observed, indicating that water forms a monolayer. These characteristics are clearly established for the first time. In the work of Laget et al., 12 the acetate character was supposed to be unidentate without experimental proof, and the role of water was not discussed. Our results indicate that LHS-Co is isostructural with zinc hydroxychloride, with the anion acetate in the tetrahedral coordination of Co and water intercalated as monolayer (Figure 10b). As for LHS-Ni, the dehydration occurred via a topotactic process.

For LHS–Zn, the Δv value remains almost constant upon dehydration, showing that the acetate anion acts as a free species intercalated between the layers. The interlamellar distance also remains almost constant upon heating. The total release of water at 110 °C led, via a destructive process, to the formation of an oxoacetate. These results indicate that LHS-Zn has a structural model closely related to the zinc hydroxynitrate form, with water occupying the fourth apex of the tetrahedron and the acetate anion intercalated between the layers (Figure 10c). Conversely, the LHS-Zn prepared by an exchange reaction¹³ conserves the interlamellar organization of the starting material $Zn_5(OH)_8Cl_2 \cdot H_2O$, which is similar to that observed for LHS-M (M = Ni and Co). This clearly evidences the influence of the preparative route on the structural characteristics of such materials.

Conclusion

This work describes a new method for preparing layered hydroxyde acetates. This novel route, involving hydrolysis and inorganic polymerization in polyol media, offers several advantages compared to classical methods conducted in aqueous solution. It allows simple control of the reaction by means of the hydrolysis ratio rather than pH as in aqueous solution, avoids contamination with carbonate anion, and permits a rapid and easy



Figure 10. Structural model proposed for (a) LHS–Ni, (b) LHS–Co, and (c) LHS–Zn.

process. The use of several techniques allows us to discuss in more detail the main characteristics of these poorly ordered compounds. Our comparative study

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revealed significant differences, which concern mainly the nature of the cation site and the organization of water molecules and acetate anions between the layers. This organization is established for the first time. LHS– Ni and LHS–Co have a reversible topotactic dehydration process, which conserves their lamellar character. This process is of interest because it shows that these two materials can be handled in a relatively wide temperature range. Conversely, LHS–Zn undergoes a destructive dehydration process leading to ZnO.

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